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# Effect of surface catalysts on the long-term performance of reactively sputtered tin and indium oxide gas sensors

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# Abstract

Long term performance of tin and indium oxide thin films with palladium and platinum deposited on their surface as catalysts was studied using a standardization method proposed recently by other researchers. Asimple gas classification algorithm was used to investigate the validity of the standardization method proposed and to study the effect of doping type and thickness onto the surface of  $SnO<sub>x</sub>$  and  $InO<sub>x</sub>$  films on their selectivity and stability characteristics. It was found that: (a) the proposed method is reliable for the assessment of sensor degradation; and (b) sensors with relatively thick additive layer are more reliable. © 1997 Elsevier Science S.A.

Keywords: Metal oxide gas sensors; Surface catalysts; Gas classification; Sensor selectivity; Sensor degradation  $\ddot{\phantom{0}}$ 

## 1. Introduction

Metal oxide gas sensors offer many advantages as simple fabrication, low cost, high sensitivity to both reducing and oxidizing gases, easy integration, etc. Main disadvantages of these sensors are poor selectivity and stability. Noble metals used as catalysts enhance their characteristics but reliable sensors have not been reported yet. It is important to fabricate sensors not necessarily sensitive to one gas, but sensors exhibiting different responses to different gases of interest ('non-selective but different'), for their incorporation in sensor arrays in order to design systems selective to the desirable gases [1,2]. Researchers are trying to apply linear regression methods like PCA (principal component analysis) or PLS (partial least squares) to enhance selectivity  $\frac{1}{2}$  of  $\frac{1}{2}$  (partial regression technology to emigric sensoring of a linear sensor analy, i withhelm regression teen niques like MARS (multivariate adaptive regression splines) [3], PPR (projection pursuit regression) [4] or TLS (transformed least squares) [5,6] are used for nonlinear sensor arrays. Even the application of linear regression arrays. Even are approached of mical regression teeninques for non-imear our requirednt sen-

parts of an array. In this work the performance of a single sensor was evaluated concerning its selectivity to four components with respect to its degradation, without considering the sensor as part of an array. This analysis can provide useful information for the characterization of a sensor, but is not adequate for the characterization of an array which would use this sensor. The performance of a sensor was modeled using the

[7]. All these methods use statistical tools to evaluate the performance of an array or individual sensor elements as

distances between different classes with respect to the translation of the sensor response in the  $n$ -dimensional space. Each class (gas) occupies a specific volume in the  $n$ -dimensional space. The pattern recognition system used is called to map the points of the sensor space asca is cancel to map the points of the sensor space. onto classification space. More than one sensors can be used, thus defining a sensor array. If the target gases do not provide well distinguished areas in sensor space, then the recognition algorithm cannot be expected to the expect of the expected to the three to the three to the traction of t discriminate the theorem. For example, if and a secure them.  $F'' = 1 - 1 - 1$ , is not possible to distinguish the gases of  $f(x) = f(x) - f(x)$ rig. Toverlap, it is not possible to distinguish the gase corresponding to these areas with a single measurement, especially if the measurement falls in the cross section of these areas. Consequently, the performance of individual sensors is of major importance for correct<br>classification.

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Fig. 1. Gas classification system.

The operation of metal oxide gas sensors is based on the interaction of ambient gases with sensor surface, This interaction leads to modification of chemical composition of the oxide and consequently to a change to its electrical properties. Since this kind of sensors operates at elevated temperatures, continuous degradation of sensor properties with operation is a usual phenomenon. The shaded regions of the hypersphere of Fig. 1 represent the regions in which the responses of a sensor to two gases A and B fall, for a number of consecutive measurements. Two measurements corresponding to gas A are shown. The area of these regions with respect to the distance between them can provide a measure of sensor reliability.

#### 2. Experimental techniques

Deposition of  $\text{SnO}_x$  and  $\text{InO}_x$  films was performed with a Leybold Z-400 planar magnetron sputtering system, which was d.c. operated in a controlled, high purity Ar-O<sub>2</sub> mixture. Two different discharge modes may be obtained, the metallic mode and the reactive mode, where the target is respectively free from or covered with reaction products. The compound formation reactions are desirable in the metallic mode for high rate depositions, and for a more precise control of the  $O_2$ /metal (Sn or In) ratio of the deposited film. This necessitates the use of a plasma emission monitor (PEM) system, that allows us to select and maintain the degree of target oxidation [8], with the tin or indium emission line intensity at 450 nm as a measurand for the control loop of the reactive gas mass flow con-

troller. This technique has been proven adequate for the fabrication of reproducible films.

Sputtering was carried out on  $\text{Al}_2\text{O}_3$  (96%) substrates with dimensions  $12 \times 26$  mm, heated at 300°C. Films were fabricated at a relatively high total gas pressure of approximately  $1.0 \times 10^{-2}$  mbar. The substrate holder was at a distance of 7 cm from the 10 cm in diameter target. The total power into the tin or indium target was 115 W with a total current of 0.36 A, thus achieving a deposition rate of approximately 100 nm min<sup>-1</sup>. Argon flow was adjusted manually at 20 ml min<sup> $-1$ </sup>, while oxygen intake was controlled by the aforementioned PEM control unit, so that the intensity of the metal (tin or indium) emission line remained constant. An  $O_2$  flow around 30 ml min<sup>-1</sup> was found to give an appropriate oxygen to metal (tin or indium) ratio, so that the films fabricated can function as sensors with a base resistance below 30 MOhms at 100°C (low oxygen to metal ratio gives metallic films and high oxygen to metal ratio gives highly resistive films). The thickness of all films fabricated was equal to 1000 nm (10 min sputtering time). Films were not annealed. The eight lots of tin oxide films, which were fabricated under the same experimental conditions, showed the same electrical properties within a 10% tolerance. In addition, the eight lots of indium oxide films, which were fabricated under the same experimental conditions, showed the same electrical properties within a 10% tolerance.

Subsequently, palladium was electron beam evaporated on three of the aforementioned  $SnO<sub>x</sub>$  films and on four of the aforementioned  $InO<sub>x</sub>$  films. Platinum was deposited on the four  $SnO<sub>x</sub>$  films and on the other four  $InO<sub>x</sub>$  films. One tin oxide film was left plain (without additive). Evaporation was performed with an Edwards 306 vacuum evaporation apparatus. Total pressure during electron beam evaporation was about  $3 \times 10^{-5}$ mbar, while the substrates were not heated during evaporation. Evaporation was carried out at a relatively low deposition rate, of the order of 3  $\AA$  s<sup>-1</sup> for palladium and 1 Å  $s^{-1}$  for platinum. Palladium was deposited simultaneously on three  $SnO<sub>x</sub>$  and three  $InO<sub>x</sub>$ films with evaporation times of 5, 10 and 20 s. On the fourth  $InO<sub>x</sub>$  film palladium deposition time was 30 s. Thus three  $SnO<sub>x</sub>$  and four  $InO<sub>x</sub>$  films with increasing Pd layer on each of them have been fabricated. We refer to these sensors as  $SnO_x + Pd1$ ,  $SnO_x + Pd2$ ,  $SnO_x + Pd3$ ,  $InO_x + Pd1$ ,  $InO_x + Pd2$ ,  $InO_x + Pd3$  and  $InO_x + Pd4$ , depending on palladium thickness. In the same manner but at a much lower rate  $(1 \text{ Å } s^{-1})$  platinum was deposited on four  $SnO<sub>x</sub>$  and four  $InO<sub>x</sub>$  films with evaporation times of 10, 20, 30 and 40 s, respectively. In an analogous sense we refer to these sensors as  $SnO_x + Pt1,..., InO_x + Pt4.$  The average thickness of the metal films deposited onto oxide surface has been evaluated by activation energy measurements, with the procedure described in [9], and has been found around 3-5 nm. For details concerning the physical properties and the structure of the additive layers one can refer to  $[9]$ .

The sensor characterization set up was designed to measure the steady-state and the transient response of the samples, testing them at various working temperatures and under different gas compositions. The testing conditions were rigorously controlled via a fully automated, computer controlled system. All sensors have been characterized according to the following experimental procedure:

- $\bullet$  The samples were heated to 450 $\degree$ C and cooled back to 150°C in zero grade air (ZGA, dry synthetic mixture of 80%  $N_2$  and 20%  $O_2$ ). This step was repeated three times before starting film characterization, in order to clean the samples from water vapor remains.
- $\bullet$  The samples were heated to 450°C and cooled back to 150°C in 2000 ppm carbon monoxide diluted in ZGA.
- $\bullet$  The samples were heated to 450 $\degree$ C and cooled back to 150°C in zero grade air.
- The last two steps were repeated four times again but with test gas compositions of 1% methane, 26 ppm ethanol and 2% butane diluted in zero grade air, instead of CO.  $\frac{1}{\sqrt{1-\frac{1$
- The samples were submitted to so thermal eyeres (heating to  $450^{\circ}$ C-cooling to  $150^{\circ}$ C), in the presence of ambient air. <sup>l</sup>The samples were tested again to all target gases,
- rile samples were tested again to all target gases, repeating the aforementioned procedure. In this way a second set of sample responses was recorded, after 50 aging cycles.

 $\bullet$  With the aforementioned procedure the response of the samples to test gases was recorded after 100, 150 and 250 subsequent aging cycles. In this way five sets of sample responses were recorded.

During the aforementioned measuring cycles, the resistance of the samples was recorded at 5°C intervals as the samples were cooled from  $450$  to  $150^{\circ}$ C; in this way 61 resistance values were recorded for the response of each sensor to each test gas. The cooling speed was  $10^{\circ}$ C/min. At temperatures above 250 $^{\circ}$ C, this speed ensures that no significant transient effects have been recorded [10].

## 3. Mathematical treatment

# $3.1.$  Pre-processing of the data

The acquired experimental data include resistance values for both tin and indium oxide based sensors, covering a range of several orders of magnitude (indium oxide based sensors are by far more conductive than tin oxide based ones; particularly platinum doped tin oxide sensors exhibit non-measurable conductance values). As a consequence, a scaling technique is necessary in order to compensate the data for large resistance variances. A simple pre-processing technique that gives each variable equal weighting by adjusting the original data to remove inadvertent weighting on values covering a wide range was adopted and it is briefly described below.

Vectors were used in order to represent the response of the sensors. Each vector contains the resistance values of the sensor in the presence of the corresponding test gas in the temperature region from  $150-450^{\circ}$ C, recorded at 5°C intervals; thus the dimension of each vector is  $61((450-150)/5+1)$ .

The original data have been autoscaled so that the vector elements of each sensor are mean centered with a standard deviation of one. The following relationship gives the formula for the calculation of the autoscaled data [11]:

$$
\tilde{r_i} = \frac{r_i - r_{av}}{\sigma}, \quad i = 1, \dots, N
$$
\n<sup>(1)</sup>

where  $r_{av}$  is the average resistance over all vector elements of each sensor at the presence of each test gas at each measuring cycle and  $\sigma$  is the standard deviation of the sample according to the following formula [11]:

$$
\sigma^2 = \frac{\sum_{i=1}^{N} (r_i - r_{\rm av})^2}{N - 1}
$$
\n(2)

where  $N = 61$  (the dimension of the vector).

 $\mathcal{N}$ 

# 3.2. Performance standardization

In order to compare the performance of sensors it is necessary to define some standards. Recently J.W. Gardner and P.N. Bartlett [l] presented a simple method to represent with a number the ability of a sensor to discriminate two gases. Their method consists of the calculation of a factor, called resolving power (or electronic selectivity), which takes into account both the distance of the response vectors to two gases and the errors encountered. In a mathematical form, the resolving power of a sensor for two gases is defined by the following equation:

$$
R = \frac{|S_{AB}|}{\sqrt{\sigma_A^2 + \sigma_B^2}}
$$
 (3)

where  $|S_{AB}|$  is the distance between in the response vectors A and B and  $\sigma_A$  and  $\sigma_B$  are the errors calculated along the direction of the distance metric. In the situation at hand,  $|S_{AB}|$  is the Euclidian distance between the responses of the sensor to two test gases and  $\sigma_A$  and  $\sigma_B$ are the distances between the responses of the sensor to the test gases A and B, respectively, before and after a predetermined number of aging cycles (these distances are encountered as errors). Referring to Fig. 1,  $\sigma_A$  and  $\sigma_B$  denote the distance of the response points of two successive measurements under test gases  $A$  and  $B$ ; the resolving power denotes the change of the distance between the two points with respect to the translation of the points with aging. Thus, if this factor is considerably larger than unity, then the sensor can discriminate the gases, while if it is smaller than unity the sensor should not distinguish the gases.

## 3.3. Classification rule

The pattern recognition engine used is a simple supervised learning algorithm. The response vectors to the four test gases are recorded during a measurement and are used as a decision rule for the successive measurement. The 'learned' response vectors are then replaced by the new measurement data. The decision rule consists of evaluating the four distances of the unknown vector from the four known classes and classifying the unknown response to the class from which this distance is minimum. Table Id, Table 2d and Table 3d show the classification results according to this simple rule. The headings show the actual gases present and the four lines under the headings show the 'class' to which the test gas was attributed. If this gas coincides with the heading the classification is correct. The first row under the heading shows the classification for the response vectors taken after 50 aging cycles, with 'training data' the response vectors taken without aging. In an analogous sense the fourth row shows the classification results for the response vectors after 250 aging cycles,

while the system was 'trained' with the response vectors after 150 aging cycles.

## 4. Results and discussion

Results from the most stable and selective sensors are presented in Tables 1-3. Tables 1-3 show the translation of the response vectors at each gas with aging (a), the resolving power for the response vectors after 100 and 150 aging cycles (b), the resolving power for the response vectors after 150 and 250 cycles (c), and the results of the gas classification algorithm (d). The headings of Table Id, Table 2d and Table 3d show the actual gas that was present. From Tables  $1-3$ , it is easy to investigate the behavior of the selectivity of each one of the sensors, in conjunction with the effect of degradation with continuous use, given that the resolving power reflects in practice sensor selectivity.

Table la shows that after 100 temperature cycles the tin oxide sensor with the thicker palladium layer becomes remarkably stable to all test gases. This stability

#### Table 1

Classification results for a palladium doped  $SnO<sub>x</sub>$  sensor

	CO	$CH_a$	C,H,OH	$C_4H_{10}$			
(a) Effect of aging							
Aging							
$0 - 50$	328	843	60	419			
$50 - 100$	357	518	41	29			
$100 - 150$	11	42	13	7			
$150 - 250$	90	26	20	79			
(b) Resolving factor, 100–150 aging cycles							
CO	0	746	1855	2667			
CH <sub>4</sub>	*	0	1167	1292			
$C_2H_3OH$	*	÷.	0	307			
(c) Resolving factor, $150-250$ aging cycles							
CO	∩	432	286	265			
$CH_{4}$	÷	0	1633	720			
C, H, OH	*	*	0	134			
(d) Classification results							
Aging							
$0 - 50$	CO.	CO.	C,H,OH	CO			
$50 - 100$	$C_4H_{10}$	$C_4H_{10}$	C,H,OH	$C_4H_{10}$			
$100 - 150$	$_{\rm CO}$	$\rm CH_{4}$	$C_2H_5OH$	$C_4H_{10}$			
150-250	CO	CH <sub>4</sub>	$C_2H_5OH$	C <sub>2</sub> H <sub>5</sub> OH			

 $T$  heading row shows the gas actually present and each row shows the gas actually present and each row shows  $\frac{1}{2}$ the nearing fow shows the gas actually present and each fow shows the output of the classification algorithm if the system has been trained with the data recorded prior 50 (rows  $1, 2$  and  $3$ ) and  $100$ (row 4) temperature cycles.  $\frac{1}{\sqrt{2}}$  Europeian distance distances to the same test vectors to t

(a) Lacinian distances among the response. gases after a fullfluer of temperature cycles.

(b) resolving factors denoting the selectivity of the sensor if it is submitted to 50 temperature cycles, with a pretreatment of 100 temperature cycles. (comperature cycles, it is submitted to 100 temperature cycles, it is submitted to 100 temperature control to 100

(c) same as  $(v)$  if the sensor is submitted to 100 with a pretreatment of 150 temperature cycles.<br>(d) Output of a classification algorithm.

Table 2 Classification results for a platinum doped  $SnO<sub>x</sub>$  sensor

	CO	$CH_{4}$	$C_2H_3OH$	$C_4H_{10}$
(a) Effect of aging				
Aging				
$0 - 50$	1363	869	706	747
$50 - 100$	1134	9	515	743
$100 - 150$	1028	25	129	146
$150 - 250$	634	8	57	133
(b) Resolving factor, 100-150 aging cycles				
CO	0	99	128	85
$CH_{4}$	×	0	722	820
C,H,OH	×.		0	685
(c) Resolving factor, $150-250$ aging cycles				
CO	n	173	208	138
$CH_{4}$	*	0	1661	923
C,H,OH	*	ж	0	951
(d) Classification results				
Aging				
$0 - 50$	C <sub>2</sub> H <sub>5</sub> OH	$C_2H_5OH$	$C_4H_{10}$	C <sub>2</sub> H <sub>5</sub> OH
$50 - 100$	$C_4H_{10}$	$CH_4$	C,H,OH	$C_4H_{10}$
$100 - 150$	$CH_{4}$	$CH_4$	$C_2H_5OH$	$C_4H_{10}$
150-250	CO	$CH_4$	$C_2H_3OH$	$C_4H_{10}$

The heading row shows the gas actually present and each row shows the output of the classification algorithm if the system has been trained with the data recorded prior 50 (rows 1, 2 and 3) and 100 (row 4) temperature cycles.

(a) Euclidian distances among the response vectors to the same test gases after a number of temperature cycles.

(b) Resolving factors denoting the selectivity of the sensor if it is submitted to 50 temperature cycles, with a pretreatment of 100 temperature cycles.

(c) Same as (b) if the sensor is submitted to 100 temperature cycles, with a pretreatment of 150 temperature cycles.

(d) Output of a classification algorithm.

results to high resolving power (shown in Table l(b, c) and consequently the sensor can distinguish the test gases with the discrimination algorithm used (Table Id). As it is obvious from Table la (first two rows) the sensor must be submitted to 100 temperature cycles in order to become stable. During this pre- treatment it is believed that severe changes take place, both to the structure and stoichiometry of the tin oxide layer, and to the form of the catalytic layer. Recall that the tin oxide film was not annealed and deposition of palladium took place at ambient temperature. Note that the sensor cannot distinguish butane from ethanol after 250 operation cycles (the recognition algorithm decided  $C_{\text{H}}$  or  $C_{\text{H}}$ , whereas  $C_{\text{H}}$ , whereas  $C_{\text{H}}$ ,  $C_{\text{H}}$ ,  $C_{\text{H}}$ ,  $C_{\text{H}}$ ,  $C_{\text{H}}$ ' $C_2H_5OH$ ', whereas ' $C_4H_{10}$ ' was present, Table 1d).<br>Note also that the relatively low resolving factor of 1.34 at Table lc implies this error.  $A = (1 - \cos k) \cdot (1 - 1 - 1)$ 

for the platform doped sensor. Note for  $\alpha$  that  $\alpha$ tor the platinum dependence. The from The sensor  $\frac{1}{2}$ ine response or the sensor to CO is not stable, whereas its response to the other test gases is stable. This can also be observed at Table  $2(b, c)$ , where the first lines (corresponding to the resolving factors of CO with the

other test gases) imply that this sensor should not recognize CO. Finally, classihcation results at Table 2d confirm the incapability of the sensor to recognize CO, whereas it can successfully distinguish the other three test gases,

Palladium doped indium oxide sensors exhibited serious degradation with temperature cycling and the classification algorithm was not able to correctly discriminate the test gases. The corresponding data are not presented.

The behavior of the platinum doped indium oxide sensor with the thicker layer of platinum is shown in Table 3. This is an example of a non-selective sensor, which although is stable (Table 3a) the low resolving factors of Table 3(b, c) imply that it is not capable to classify correctly the test gases. This happens indeed, as shown in Table 3d. Indium oxide based sensors were generally found non-selective. Their behavior continues to degrade even after they were submitted to 250 operation cycles.

The structure of palladium and platinum films deposited on tin and indium oxide films was estimated by

#### Table 3

Classification results for a platinum doped  $InO<sub>x</sub>$  sensor

	CO	CH <sub>4</sub>	$C_2H_3OH$	$C_4H_{10}$			
(a) Effect of aging							
Aging							
$-0-50$	1158	1539	1344	1051			
$50 - 100$	229	223	735	247			
$100 - 150$	172	170	151	104			
$150 - 250$	136	149	48	208			
(b) Resolving factor, $100-150$ aging cycles							
CO	0	61	58	194			
CH <sub>4</sub>	*	0	82	123			
C <sub>2</sub> H <sub>5</sub> OH	*	$\ast$	0	160			
	(c) Resolving factor, 150–250 aging cycles						
CO	0	45	58	194			
CH <sub>a</sub>	*	0	95	205			
$C_2H_3OH$	$\ast$	*	0	218			
(d) Classification results							
Aging							
$0 - 50$	C, H, OH	C, H, OH	C,H,OH	$C_2H_5OH$			
$50 - 100$	CO	CO.	CO	CH <sub>a</sub>			
$100 - 150$	CO.	CO.	$\rm C_2H_5OH$	$C_4H_{10}$			
$150 - 250$	$\rm CH_{4}$	CH <sub>4</sub>	CO	$C_4H_{10}$			

The heading row shows the gas actually present and each row shows the gas actually present and each row shows  $\alpha$ the heading fow shows the gas actually present and each fow shows the output of the classification algorithm if the system has been trained with the data recorded prior 50 (rows 1, 2 and 3) and  $100$  (row 4) temperature cycles.  $(10w + j \text{ temperature cycles})$ 

(a) Euchdian distances among the response. gases after a number of temperature cycles.

(b) Resolving factors denoting the selectivity of the sensor if it is submitted to 50 temperature cycles, with a pretreatment of 100 temperature cycles.  $m_{\text{eff}}$  is submitted to sensor is submitted to  $100$  temperature cycles,

(c) same as  $(0)$  if the sensor is submitted to 100 with a pretreatment of 150 temperature cycles.<br>(d) Output of a classification algorithm.

activation energy measurements, with the use of conductance versus temperature plots. It was found that the thicker palladium (Pd3) and platinum (Pt4) films (average thickness  $> 30$  Å) consist of discrete islands with particle dimension around  $80 \text{ Å}$  and particle separation in the order of a few tenths  $\AA$  [12-14]. It is also known that when thin discontinuous metal films are subjected to high temperatures small metal islands grow together to form particles with larger dimensions [12]. This is a degradation mechanism for the sensors, which has a larger effect for thin catalyst films. After some thermal pretreatment this mechanism stops and does not contribute to sensor degradation. Films covered with thin catalyst layers showed worse sensing properties than the ones shown in Tables 1-3, and their response is not illustrated.

#### 5. Advantages of the proposed method

The mathematical method proposed makes a compression of the information contained in the conductivity measurements of the sensors. Information presented in Tables l-3 would require many conductivity graphs to be displayed, the study of which would be cumbersome and misleading. The most useful characteristics of the sensors, i.e. stability or selectivity, would be very difficult to be extracted from conductivity measurements. On the other hand, the tables with the resolving factors presented give an overview of the reliability of the sensors, taking into account both their selectivity and stability.

However, the classes to which the responses of the sensors were classified refer to single concentrations of five reducing gases. Similar results would presumably be obtained if different gas concentrations were used, but if the sensors were exposed to gas mixtures the results would be very different, since synergetic effects would occur [15]. Hence, the results of this mathematical method can be applicable if the gas concentrations are different. The proposed method is not suitable for the processing of data corresponding to the exposure of sensors to arbitrary gas mixtures.

#### 6. Conclusions

The performance of tin and indium oxide sensors, doped with varying thicknesses of palladium and platinum was evaluated against accelerated temperature aging, using a simple mathematical formula proposed refines, doing a omipic maintmantar romina proposed recently by other researchers, this formula was proven adequate to correctly describe their performance. It was found that the thickness of the additive plays an important role in sensor stability and its capability to discriminate the four gases tested. A simple supervised pattern

recognition engine was used in order to assess the selectivity of the sensors. Sensors covered with very thin noble metal layers were found unreliable, while sensors covered with a thicker catalyst layer were found more stable and selective. Tin oxide based sensors were found more reliable than indium oxide based ones, for the temperature treatment they were submitted to.

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## **Biographies**

 $J_{\text{A}}$  and  $J_{\text{B}}$  in  $J_{\text{B}}$  in  $J_{\text{B}}$  in 1948. He  $\frac{1}{2}$ received his B.Sc. (Hon.) in physics from the Depart-<br>ment of Physics of the University of Athens in 1972. His M.Sc. and Ph.D. degrees were obtained from Loughborough University of Technology, UK, in 1974 and 1976, respectively, in the field of thin-film technology and fabrication of thin-film devices.

From 1976 to 1979 he was employed as a research fellow in the Thin Film Group of Loughborough University of Technology. From 1980 to 1986 he was appointed lecturer at the Electronics Laboratory of Athens University, and in 1986 he was elected associate professor in the Department of Electrical Engineering of the National Technical University of Athens. In 1990 he was elected professor of microelectronics in the same department.

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